ASSESSMENT OF THE INTEGRITY OF PIPELINES SUBJECT TO CORROSION-FATIGUE, PITTING CORROSION, CREEP AND STRESS CORROSSION CARCKING

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## **Presentation Outline**

- Background
- Objectives
- Failure Mechanism Modeling and Applications
  - Corrosion-Fatigue
  - Pitting Corrosion
  - Creep
  - Stress Corrosion Cracking
- Conclusions

# Objectives

- Develop Physics-based computationally <u>simple</u> probabilistic models for routine reliability assessments and health monitoring in the oil industry
  - PoF (Physics of Failure) models capture material degradation and failure mechanism and can be extrapolated to different levels.
  - Probabilistic models can adequately represent all of the factors that contribute to variability (e.g. material properties, Inspection devices accuracy, human errors, etc.)
  - Use of the probabilistic models to estimate reliability of components (our interest is pipeline reliability)

### Problem Statement



# **Corrosion-Fatigue in Pipes**





# **Corrosion-Fatigue Modeling Approach**



### Damage-Endurance Reliability Models



### Probabilistic Fracture Mechanics Approach to Fatigue Reliability



### Damage-Endurance Modeling Corrosion-Fatigue



### Physics-Based Simulation Results



### Semi-Empirical Simplified Model Development



Find the correlation of A & B with the physical parameters of the pipeline:

- Loading Stress "σ"
- Loading Frequency "v"
- Temperature "T"
- Flow Characteristic "C" (e.g., Ip, [Cl<sup>-</sup>], pH, ...)

*Damage*,  $D = f(v_i, t_i | \Theta)$ 

 $D \approx$  e.g. crack size, a

- $v_i \approx \text{variables (e.g. T, <math>\sigma, \nu, [\text{Cl}^-], ...)$
- $t_i \approx \text{index of time (e.g. N,...)}$
- $\Theta \approx$  vector of model constants (e.g.  $\varepsilon_i, A, B, ...$ )

### Proposed Simple PoF-Based Corrosion-Fatigue Semi-Empirical Model



# Data Collection: Experimental Validation



# **Cortest Corrosion-Fatigue Testing Results**

- Crack length vs. number of cycle from Cortest corrosion-fatigue testing:
  - in sea water of 250 ppm  $Na_2S_2O_3$  -5H<sub>2</sub>O at 383 K, and 100 and 150-MPa,
  - at different frequencies of 0.004 Hz and 0.00165 Hz.
  - Pictures from broken specimen with connected screws(for applied current and voltage) are shown at the bottom.



# Broken specimen from Cortest corrosion-fatigue testing

• Broken specimen (two side views plus broken surface parts)







### voltage -

#### current - connections

### Model Parameter Estimation



### Reliability and Health Monitoring Application



# Reliability and Health Monitoring Application (Cont.)



# Pitting Corrosion (Phase II)

 With Collaboration from PI Summer Interns Abdullah M. Al Tamimi & Mohammed Mousa Mohamed Abu Daqa

# Background

#### **Pitting Corrosion (X70Carbon Steel)**

- Pitting Corrosion: An electrochemical oxidation reduction process, which occurs within localized holes on the surface of metals coated with a passive film.
- It might be accelerated by chloride, sulphate or bromide ions in the electrolyte solution.
- Pitting corrosion has a great impact on the oil and gas industry.
- There are three main stages for the pitting corrosion to occur:





# **Objectives of Pitting Corrosion**

#### • Objectives

- 1. Measuring pits depth,
- 2. Measuring pits density, and
- 3. Measuring the mass loss.
- Two Corroding Environments (X70 Carbon Steel at 323 K) :

-  $H_2S = Na_2S_2O_3-5H_2O$  with 100 100ppm, 150ppm, 200ppm, 300ppm and 400ppm concentration, in 5,10, 24 hours time period;

- Chloride(Sea Water) with 100 100ppm, 150ppm, 200ppm, 300ppm and 400ppm concentration, in 5, 10, 24 hours time period.

# Experimental Setup/1

• The scheme of the experimental setup:



# Experimental Setup/2

• The scheme of the experimental setup:



# Static stress corrosion specimen with a strain gage on it to measure the applied stress.

# Examples of Pitting Corrosion H<sub>2</sub>S and Chloride-Results

- Morphology of the samples are studied by:
  - 1- Optical Microscope, Nikon Optiphot 66
  - 2- Sensitive Weighing Machine, METTLER TOLEDO AB104
  - 3- Scanning Electron Microscope, HITACHI SU-70 SEM
- Morphology of Pits on X70 Carbon Steel surface in corrosive environments



in  $H_2S$  (Mx200)



### Pitting Depth of X70 Carbon Steel (H<sub>2</sub>S and Chloride-Results)

• Pit depths for unstressed samples (left for H2S, right for Chloride) followed Weibull distributions.  $D_{H.S} \sim Weib(\beta = 6.55, \alpha = 11.99 \mu m)$   $D_{[Cl^-]} \sim Weib(\beta = 1.32, \alpha = 57.68 \mu m)$ 



#### Pit Depth Distribution 400 ppm

## Pitting Density of X70 Carbon Steel (H<sub>2</sub>S and Chloride-Results)

• Pit densities followed the lognormal distributions.

 $PD_{[h_2S]} \sim LN(\mu = 3.06 , \sigma = 0.39)$ 

$$PD_{[Cl^{-}]} \sim LN(\mu = 2.18 , \sigma = 0.30)$$



 Pit Density distribution 400ppm, 5hours are given in [pits/cm<sup>2</sup>] The actual mean: [8 pits/cm<sup>2</sup>] (left), [9 pits/cm<sup>2</sup>] right

### Estimation of Pitting Corrosion Characteristics (stressed and unstressed)

- 250 ppm  $H_2S$  (Sodium-thio-sulfate) at 80°C (353K).
- Mean Intensity: 14 in 250x250  $\mu$ m<sup>2</sup> (0.0625 mm<sup>2</sup>).
- The lognormal plotting diagrams of the unstressed and stressed

99.00

50 00

10.00

5.00

1:00

10.00







Probability Plot

Number of Pits

100.00

### Pit Growth Rate Model

 Pit depths (d) increases with the concentrations according to a power law and time according to the t<sup>1/3</sup>-law (justified by the literature ): d = A t<sup>m</sup>



# **Creep Modeling Background**

- Creep is the time-dependent, thermally assisted deformation of materials under constant static load (stress).
- Mathematical description of the process is difficult and is in the form.

$$\varepsilon = f(\sigma, T, t)$$

• Creep at low temperatures (primary stage) are described by:

$$\varepsilon = \varepsilon_0 + \alpha \log(1 + \gamma t)$$

$$\varepsilon = \varepsilon_0 \beta t^{1/3}$$

- where  $\alpha$ ,  $\beta$  and  $\gamma$  are material constants;
- There is no general agreement on the form of the equations at high temperatures.

# **Creep Modeling Background (Cont.)**

- A typical creep curve shows three distinct stages with different creep rates, determined by several competing mechanisms from:
  - strain hardening,
  - softening processes such as recovery and crystallization,
  - damage processes such as cavitation, necking and cracking.



 Creep testing and the creep curve, showing how strain ε increases with time t up to the fracture time. [http://faculty.mercer.edu/bubacz\_m/Links/CH13.pdf]

### **Creep Approach Modeling**

- Only literature search completed with some preliminary experimental preparations. The approach includes
  - 1. Using simulation of detailed models propose an empirical model.
  - 2. Perform accelerated creep tests.
  - 3. Use experimental results to assess parameters and uncertainties of the proposed empirical model accelerated life testing.

## Creep Accelerated Test Set up

• The creep test is carried out by applying a constant load to a tensile specimen maintained at a constant temperature, (according to ASTM E139-70).





[http://www.sut.ac.th/engineering/Metal/pdf/MechMet]

### **Creep and SCC Experimental Setup**

- Two chambers designed for creep and SCC tests under different environmental conditions and applied stress:
- Left: chamber for Dog-bone
- Right: chamber for CT-specimen (The prototypes specimens at work, installed on MTS-machine).





Dog-bone --specimen

#### **CT-specimen**

### **Stress Corrosion Cracking (SCC)**

- SCC is a combination of static tensile stress below yield and corrosive environment.
- Tensile stresses may be external forces, thermal stresses, or residual stresses.
- The kinetics of SCC depends on three necessary conditions:
- 1. The chemical and metallurgical state of the material (chemical composition , thermal conditions, grain size, presence of secondary phases and precipitate, etc.)
- 2. The environmental conditions (environmental composition, temperature, pressure, pH, electrochemical potential, solution viscosity etc.)
- 3. Stress state (uniaxial, triaxial, etc.) and on crack geometry of the material.



### **Stress Corrosion cracking(Factor Affecting)**

- General relationship for the penetration of SCC following commonly accepted dependencies (after Staehle).
- There are many submodes of SCC and, because of the large number of variables in Staehle's equation, there is a great range of possibilities in the study of SCC. This contributes to the complexity of the subject

# $X = A [H^{+}]^{n} . [x]^{p} . \sigma^{m} . e^{(E - E_{0} / b)} . e^{Q / RT} . t^{q}$

- Where X is the depth of SCC penetration;
- A depends on alloy composition and structure;
- $[H^+]$  is PH; x is the environmental species;
- $\sigma$  is stress;
- E is electrochemical potential;
- Q is the activation energy;
- R is gas constant; T is temperature;
- t is time;
- n, p, m, b, q are empirical constant

[Kenneth R. Trethewey; Materials & Design; Volume 29, Issue 2, 2008, Pages 501-507]

### **SCC Planned Tests**

• Tests on statically loaded (stressed) smooth specimens



## Planned Tests on statically loaded pre-cracked samples

- Fracture mechanics testing for SCC conducted with either :
  - a constant load or
  - with a fixed crack opening displacement,
- the *da/dt* is measured.
- The crack depth is determined as a function of time and the stress intensity.
- $K_{1SCC}$  is the min. stress intensity below which SCC does not occur.



CT- specimen for fracture- mechanic-type testing where crack velocity vs. stress intensity is obtained



Schematic plot of data from fracture- fracture-type Testing.  $K_{ISCC}$  is shown to

# Conclusions

- Reliability models for Corrosion-Fatigue has been developed, verified and demonstrated
- Pitting corrosion is nearly completed with models developed for pitting depth and density
- Literature search for creep models is completed, model developed and validation to follow
- SCC modeling will start in the future--Preliminary test planning is performed